

Synthesis of BaTiO₃ Nanowires via Anodic Aluminum Oxide Template Method Assisted by Vacuum-and-Drop Loading

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Citation: Aisu K, Osada M, Suzuki Y (2014) Synthesis of BaTiO₃ Nanowires via Anodic Aluminum Oxide Template Method Assisted by Vacuum-and-Drop Loading. J Mater Sci Nanotechnol 1(1): S101. doi: 10.15744/2348-9812.1.S101

Received Date: March 28, 2014 Accepted Date: April 28, 2014 Published Date: April 30, 2014

Abstract

In this paper, we report on the synthesis of BaTiO₃ nanowires via the anodic aluminum oxide template method. To fill in the precursors of BaTiO₃ into anodic aluminum oxide templates, the vacuum and drop loading method developed in our previous study was used. Ba(CH₃COO)₂ (barium acetate) and $C_{12}H_{28}O4Ti$ (tetraisopropyl orthotitanate) were used as Ba and Ti sources, respectively. Anodic aluminum oxide membranes with the through-hole diameter of ~200 nm were used as the template for BaTiO₃ nanowires. The sample calcined at 700 °C for 2 h with NaOH post treatment at 40 °C for 72 h to remove the anodic aluminum oxide was substantially composed of BaTiO₃. Preliminary results on dielectric measurements were also discussed.

Keywords: BaTiO₃; Nanowire; Anodic aluminum oxide; Tetraisopropyl orthotitanate; Vacuum-and-drop loading method; Dielectric constant

Introduction

In recent years, miniaturization of sensors, actuators, capacitors and micro-electro-mechanical systems makes rapid progress. In order to realize the further miniaturization, nanostructured piezoelectric ceramics have been intensively studied. By controlling their nanostructures, one can make them to exhibit phase transition behavior and dielectric constants different from those of the bulk, partially due to the particle size effect [1-4]. To date, besides equiaxed nanoparticles and nanoceramics, one-dimensional (1-D) ferroelectric and piezoelectric nanomaterials are of great concern [5-9].

A variety of methods have been reported to synthesize 1-D nanomaterials. For oxide-based nanomaterials, three methods are typically used, viz., (1) hydrothermal method (using anisotropic crystal growth) [10,11], (2) anodic oxidation method (from metal precursors) [12] and (3) anodic aluminum oxide (AAO) template method [13-16]. The AAO-template method is a synthesis process in which AAO film with uniform 1-D pore channels is used as a template for 1-D nanostructure [17,18]. Due to its versatile applicability, we have focused on the AAO template method to prepare TiO₂ nanorods and nanowires [13]. Just by changing the hole diameter and thickness of AAO (or penetration depth into AAO), size and aspect ratio of 1-D nanomaterials can be easily controlled. In addition, 2-D arrays of 1-D nanomaterials can be synthesized through the appropriate removal of the AAO template.

After the Restriction of Hazardous Substances Directive by the European Union in 2003, development of lead free piezoelectric materials has been actively conducted. Despite its relatively low Curie temperature, BaTiO₃ is one of the potential candidates as lead free piezoelectric materials in some specific applications [19,20]. Hence, in this paper, we report the synthesis of BaTiO₃ nanowires via the AAO template method. To fill in the precursors of BaTiO₃ into AAO templates, the vacuum-and-drop loading method (VDLM) developed in our previous study was used [13].

Experimental

Materials

 $Ba(CH_3COO)_2$ (barium acetate, 99.8%, Lot No. WEF3058, Wako Pure Chemical Ind. Ltd., Osaka Japan) and $C_{12}H_{28}O_4Ti$ (tetraisopropyl orthotitanate, TTIP, Lot No. T88NH, Tokyo Chemical Ind. Co. Ltd., Tokyo, Japan) were used as Ba and Ti sources, respectively. CH_3COOH (acetic acid, 99.9%, Lot. No. TLE2561, Wako Pure Chemical) and $C_5H_8O_2$ (acetylacetone, ACA, 99.9%, Lot No. TLN0469, Wako Pure Chemical) were used as solvent and chelating agent, respectively. The addition of ACA is effective to slow down the hydrolysis reaction of TTIP [6,16]. For the following sample preparation procedure, $Ba(CH_3COO)_2$: TTIP : ACA = 1:1:1 in molar ratio, and $(Ba(CH_3COO)_2 + TTIP) : (CH_3COOH + ACA) = 1:4$ in volume ratio were used.

Sample preparation

At first, $Ba(CH_3COO)_2$ was put into a beaker and then CH_3COOH was poured to it. The $Ba(CH_3COO)_2$ - CH_3COOH solution was stirred with a magnetic stirrer for 90 min. TTIP and ACA were drop wisely added to the Ba source solution, resulting into a yellowish solution. The solution was further stirred for 30 min. AAO membranes (Anodisc13, Whatman) are used as the template for BaTiO₃ nanowires. The macroscopic diameter and thickness of AAO were 13mm and 60mm and the through-hole diameter was ~200 nm. AAO was dried at 100 °C for 1 h to remove adsorbed water. To fill in the precursors of BaTiO₃ into AAO templates, the VDLM was used (Figure 1) [13]. In short, the AAO template was put on the filtering flask as a cover plate. The precursor solution was repeated for 3 times. After inserting the precursor solution, AAO membranes was left in the air for 30 min to let precursor solution (i.e. TTIP in it) hydrolyze. Then, the samples were calcined at either 600, 700, 800 or 900 °C for 2 h in air. In order to remove the AAO template, the samples after calcination were put in a centrifuge tube filled with 1 M NaOH aq., and they were mildly heat treated at 40 °C for 72 h. After the NaOH treatment, samples were separated into solid phase (nanowires) and liquid phase by centrifugation (10000 rpm).



Figure 1: Schematic illustration of vacuum-and-drop loading method (VDLM). Reproduced with permission from [13]. Copyright 2013, The Ceramic Society of Japan.

Characterization

Crystal structure of samples was identified by X-ray diffraction (XRD, 40 kV, 40 mA, Multiflex Rigaku, Tokyo, Japan). A reflection free Si sample holder was used for the XRD measurement. The microstructure of nanowires was observed by scanning electron microscopy (SEM, JSM-6500F, JEOL, Tokyo, Japan). Au was coated for SEM analysis to avoid charging, typically ~5 nm in thickness. Energy dispersive X-ray spectrometry (EDS) was used for the elemental analysis.

The dielectric properties of the samples were investigated by impedance measurements, which were performed with Au/BaTiO₃/Pt micro capacitors using microprobe technique [21,22]. The samples were dispersed onto a Si substrate coated with 100 nm thick Pt film. 20 μ m Au dots were deposited on a large BaTiO₃ aggregate by e-beam evaporation through a shadow mask. The dielectric measurements were carried out in a frequency range from 100 Hz to 4 MHz and at an oscillation voltage of 0.1 V using an impedance analyzer (Agilent 4294A). For comparison, complimentary data were obtained from a fine commercial BaTiO₃ powder (BT-01, S_{BFT}=12.5 m²/g, ~ 100 nm, Ba/Ti=0.99 in atomic ratio, Sakai Chemical Ind. Co. Ltd., Osaka, Japan)

Results and discussion

XRD analysis

Figures 2(a) and 2(b) show XRD patterns for the samples calcined at 600-900 °C before and after the NaOH treatment, respectively. Without the NaOH treatment (Figure 2(a)), XRD peaks from BaTiO₃ were confirmed for the samples calcined at 700-900 °C. In addition, amorphous peaks from the AAO template were found for the samples calcined at 600-800 °C, whereas crystalline Al_2O_3 peaks were obtained for the sample at 900 °C. With the NaOH treatment (Figure 2(b)), XRD peaks from BaTiO₃ were confirmed for all samples. Instead of amorphous peaks from the AAO template, some $Al(OH)_3$ peaks were found for the samples calcined at 600-800 °C. For the sample at 900 °C, crystalline Al_2O_3 peaks remained. These results strongly suggest that the amorphous AAO template can be dissolved by the NaOH treatment (at 40 °C for 72 h), however, some remnant Al^{3+} ions (originated from the AAO template) tend to convert into $Al(OH)_3$, particularly observed for the 600 °C and 800 °C samples. The formation of $Al(OH)_3$ can be deduced as follows. During the NaOH aq. treatment, AAO templates (viz., amorphous Al_2O_3) reacted with NaOH aq. to form $Na[Al(OH)_4]$ complex. Even after washing with water and ethanol, some $Na[Al(OH)_4]$ remained on/within the samples. Then, it decomposed into $Al(OH)_3$ by the reaction with atmospheric CO_3 .



Figure 2: XRD patterns of the samples calcined at 600-900 °C: (a) without NaOH post-treatment and (b) with NaOH post-treatment at 40 °C for 72 h

The XRD patterns indicate that the formed $BaTiO_3$, before and after the NaOH treatment, seems to be composed of cubic phase (or pseudocubic). However, it is not so clear due to the limited quantity of the samples. The XRD pattern for the sample calcined at 600 °C followed by the NaOH treatment (the bottom pattern of Figure 2(b)) demonstrated that the calcination at 600 °C for 2 h was insufficient to obtain well-crystallized $BaTiO_3$.

Microstructure

Figure 3 shows SEM images of the samples calcined at 600-900 °C with NaOH post-treatment. For the 600 °C, 700 °C and 800 °C samples, aggregated bundles of BaTiO₃ nanowires were observed. The diameter of each nanowire was ~200 nm, in good agreement with the hole diameter of AAO template. Breakage of nanowires is attributable to insufficient filling of the precursor liquid and the shrinkage during the calcination process. For the 900 °C sample, amorphous AAO template converted into crystalline Al_2O_3 as shown in Figure 2, which could not be removed by the NaOH post-treatment. Thus, crystalized templates (with $BaTiO_3$) keeping the initial layer-like structure were observed.



Figure 3: SEM images of the samples calcined at 600-900 °C with NaOH post-treatment at 40 °C for 72 h

Figure 4 shows the results of EDS analysis for the samples calcined at 600-800 $^{\circ}$ C with NaOH post-treatment at 40 $^{\circ}$ C for 72 h. In good agreement with XRD results, the 600 $^{\circ}$ C and 800 $^{\circ}$ C samples contained relatively large Al peaks, whereas the 700 $^{\circ}$ C sample was substantially composed of BaTiO₃.



Figure 4: EDS spectra of the samples calcined at 600-800 $^{\circ}$ C with NaOH post-treatment at 40 $^{\circ}$ C for 72 h. Au peaks are from surface coating. The inserts almost correspond to the analyzed areas

These structures may exhibit distinct properties from bulk due to their small size and large surface to volume ratio. Dielectric permittivity (ε_r) and loss tangent (tan δ) as a function of frequency at room temperature for the sample calcined at 800 °C are shown in Figure 5. Difficulties of device fabrication do not allow us to measure 700 °C sample with pure BaTiO₃ phase. In the 800 °C sample, the dielectric response includes some contributions from Al(OH)₃ impurity and the data should thus be viewed with some caution. Nevertheless, the 800 °C sample showed a stable dielectric response ($\varepsilon_r > 800$) over a wide frequency region, which was almost comparable with well controlled commercial BaTiO₃ nanoparticles. Thus, even with some intrinsic degradation arising from Al(OH)₃ impurity, we expect that additional factors may compensate and/or enhance the dielectric response for BaTiO₃ nanowires. Since the diameter of nanowires can be tuned by changing the AAO templates, much higher dielectric properties may be obtained by using the particle size effect [1]. Such studies are underway.



Figure 5: The dielectric properties of the sample calcined at 800 $^{\circ}$ C with NaOH post-treatment, viz., BaTiO₃ nanowire (+ Al(OH)₃ "composite"). Reference sample was commercial BaTiO₃ nanoparticles.

Conclusions

We have synthesized BaTiO₃ nanowires via vacuum-and-drop method using AAO templates. Bundles of BaTiO₃ nanowires with the diameter of ~200 nm were successfully obtained. The sample calcined at 700 °C for 2 h with NaOH post-treatment at 40 °C for 72 h was substantially composed of BaTiO₃. Preliminary dielectric measurement for the sample calcined at 800 °C, even with some Al(OH)₃ impurity, has suggested the future possibility of BaTiO₃ nanowires as high performance materials. More efforts to obtain well-defined pure BaTiO₃ nanowires and to evaluate them are on-going.

Acknowledgments

This work was supported by Grant-in-Aid for Challenging Exploratory Research (No. 23655197). The authors thank Dr. Tohru S. Suzuki for the use of SEM. We also appreciate Sakai Chemical Ind. Co. Ltd for the supply of BaTiO₃ reference nanopowder.

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